Simulating study of atmospheric corrosion of steels in a coastal industrial zone: Effect of $SO_3^{2^-}$ on the formation of β -FeOOH rust particles synthesized from FeCl₃ solutions

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Abstract

For simulation of the atmospheric corrosion of steel in a coastal industrial district, β -FeOOH rusts were synthesized by aging the aqueous FeCl₃ solutions containing SO₃²⁻. The yield and crystallite size of β -FeOOH were remarkably decreased by adding SO₃²⁻, implying the suppression of formation and crystallization of β -FeOOH. Before aging, added SO₃²⁻ broke the coordination of Cl⁻ and OH⁻ to Fe³⁺ in Fe³⁺-complexes and reduced the Fe³⁺ to Fe²⁺. These facts suppose that SO₃²⁻ generated by dissolution of SO_X gas in thin film water on the steels at a coastal industrial zone markedly inhibits the formation of β -FeOOH rust.

Keywords: A. Steel; A. Iron; B. X-ray diffraction; C. Atmospheric corrosion; C. Rust

1. Introduction

Atmospheric corrosion of steels forms various kinds of steel rusts such as iron oxyhydroxides (α -, β - and γ -FeOOH), Fe₃O₄, poorly crystallized iron oxides and so forth. Among them, structure and composition of FeOOH rusts are strongly dependent of the exposure environment of steel [1-4]. The β -FeOOH rust is formed in Cl⁻-containing environment such as coastal and marine zones [1-2]. While, the α - and γ -FeOOH rusts are generated in urban and industrial zones including SO_x and NO_x in atmosphere [4]. This difference can be explained by considering that the Cl⁻, SO_4^{2-} and NO_3^{-} respectively produced by dissolution of air-borne salt, SO_X and NO_X in thin film water on the steels markedly affect on the formation and composition of steel rusts [1-3]. Therefore, studying the influence of anions on the formation and structure of steel rusts is necessary to understand the atmospheric corrosion mechanism of steels. In particular, investigation of formation of β-FeOOH rust in the presence of anions generated from SO_X and NO_X is considerably important because major industrial districts of the world are developed at a coastal area. It is well-known that major components of SO_x and NO_x are SO_2 and NO_2 , respectively, and dissolution of these gases in thin film water on the steels leads to the following reactions [3]:

The corrosion behavior and rust formation of steel have been examined by exposure test in marine atmosphere containing SO₂ [5-12]. However, a little study has been done about the influence of SO₄²⁻ and NO₃⁻ on the formation of β -FeOOH rust [13-21]. Oh et al. indicated by preparation of rust particles in a mixture of FeCl₂ and FeSO₄ solutions that increasing molar ratio [SO₄²⁻]/[Cl⁻]

enhances the α-FeOOH formation [14]. Ishikawa et al. and Kamimura et al. reported that the formation and crystallization of β-FeOOH particles are suppressed by addition of SO₄²⁻ and this behavior is associated with the stability of coordination of anions to Fe³⁺ in Fe³⁺-complexes [16,17]. Also, the authors synthesized the steel rust particles in a mixture of aqueous FeCl₃, Fe₂(SO₄)₃ and Fe(NO₃)₃ solutions and revealed that the SO₄²⁻ markedly inhibits the β-FeOOH formation, while no remarkable effect of NO₃⁻ is recognized [19]. Similar results were found for the steel rust particles prepared by aerial oxidation of a mixture of aqueous FeCl₂, FeSO₄ and NaNO₃ solutions [20,21]. On the other hand, sulfite ions (SO₃²⁻) generated by reaction (2) have almost not received attention in rust formation because the SO₃²⁻ is finally oxidized to SO₄²⁻ via reaction (3). Nonetheless, since the stability of coordination of SO₃²⁻ to Fe³⁺ in Fe³⁺-complexes is higher than that of SO₄²⁻ [22], SO₃²⁻ would affect on the formation of β-FeOOH rust as well as SO₄²⁻.

The aim of this study was to clarify the effect of SO_3^{2-} on the formation and structure of β -FeOOH rust. So that, the β -FeOOH particles were synthesized by aging the aqueous FeCl₃ solutions containing different amounts of Na₂SO₃, and the formation and structure of the products were examined by various means. The results obtained must serve to elucidate the atmospheric corrosion mechanism of steels in a coastal industrial zone.

2. Experimental

2.1 Synthesis of β -FeOOH particles in the presence of SO₃²⁻

The β -FeOOH particles were synthesized by hydrolysis of aqueous FeCl₃ solutions as follows. Aqueous solutions (250 mL) dissolving 0.03 mol of FeCl₃ were prepared in a sealed polypropylene vessel. To the FeCl₃ solutions, aqueous solutions (50 mL) dissolving various amounts of Na₂SO₃ were added.

Then, molar ratio $SO_3^{2^-}/Fe^{3^+}$ in the solution ranged from 0 to 0.4 and the Fe^{3^+} concentration was 100 mmol/L. The solutions thus prepared were aged at 85°C for 24 h in an air oven without stirring. The resulting precipitates were filtered off using 0.45 µm Millipore filter, fully washed with deionized-distilled water and finally dried in an air oven at 50°C for 24 h. All the chemicals purchased from Wako Pure Chemical Co. were regent grade and used without further purification.

2.2 Characterization

The samples thus obtained were characterized by a variety of conventional techniques. Powder X-ray diffraction (XRD) patterns were taken by a Rigaku diffractometer with a Ni-filtered Cu*K* α radiation at 30 kV and 15 mA. Particle morphology was observed by a TOPCON transmission electron microscope (TEM) at 200 kV. UV spectra were measured by a Shimadzu UV-vis spectrometer and quartz cell. Fe²⁺ concentration in the solution was assayed by an *o*-phenanthroline color comparison method.

3. Results and Discussion

Fig. 1 plots yield of the product as a function of molar ratio $SO_3^{2^-}/Fe^{3^+}$. It is clearly seen that yield of the product linearly decreases with the increase of $SO_3^{2^-}/Fe^{3^+}$ ratio and reaches to zero at $SO_3^{2^-}/Fe^{3^+} \ge 0.3$. This fact indicates that added $SO_3^{2^-}$ strongly suppresses the formation of β -FeOOH rusts.

Fig. 2 shows the XRD patterns of the products prepared at different SO_3^{2-}/Fe^{3+} ratios. At $SO_3^{2-}/Fe^{3+} = 0$, the diffraction peaks due to β -FeOOH (PDF no. 34-1266) mainly develop at $2\theta = 11.8^{\circ}$, 16.8° , 26.6° , 34.1° , 35.1° , 39.0° , 46.3° and 55.9° , corresponding to the reflection from (101), (200), (301), (004), (211), (310), (114) and (215) planes of the crystal (pattern **a**). Besides, the peaks

related to secondary phase are not detected. On raising SO₃²⁻/Fe³⁺ ratio, the peaks characteristics of β-FeOOH are gradually weakened and broadened, while the position of the peaks is unchanged (patterns **b** - **f**). Furthermore, weak peaks due to α -FeOOH (no. 81-464) appear at SO₃²⁻/Fe³⁺ = 0.15 and are intensified with an increment of SO₃²⁻/Fe³⁺ ratio (patterns **d** - **f**). Fig. 3 plots the crystallite size of β-FeOOH evaluated from full width at half maximum of the (301) peak at 2θ = 26.6° using the Scherrer equation against molar ratio SO₃²⁻/Fe³⁺. The crystallize size is 106 nm at SO₃²⁻/Fe³⁺ = 0 and is dramatically decreased by raising SO₃²⁻/Fe³⁺ ratio. From these results, it is most likely that formation and crystallization of β-FeOOH are markedly impeded by addition of SO₃²⁻.

Fig. 4 shows the TEM pictures of the particles synthesized at various molar ratios $SO_3^{2^-}/Fe^{3^+}$. The rod-shaped β -FeOOH particles are observed at $SO_3^{2^-}/Fe^{3^+}$ = 0 and their mean particle length and width are 278 nm and 61 nm, respectively (picture **a**). The size of β -FeOOH particles slightly decreases as $SO_3^{2^-}/Fe^{3^+}$ ratio increases to 0.1, meaning the suppression of growth of the particles by added $SO_3^{2^-}$ (pictures **b** and **c**). At $SO_3^{2^-}/Fe^{3^+} \ge 0.15$, the size of β -FeOOH particles increases and the irregular particles with a size of ca. 20 nm are formed (pictures **d** - **f**). The irregular particles would be identified as α -FeOOH because the formation of this material is recognized at the same $SO_3^{2^-}/Fe^{3^+}$ ratio by XRD measurements shown in Fig. 2.

The foregoing results indicate that the SO_3^{2-} generated by dissolution of SO_X gas in thin film water on the steels at a coastal industrial atmosphere remarkably inhibits the formation and crystallization of β -FeOOH rust.

It is well-known that formation of iron oxyhydroxide particles such as β -FeOOH is dependent of the solution pH [1,4,23,24]. Hence, pH of the solution before aging is plotted as a function of SO₃²⁻/Fe³⁺ ratio in Fig. 5. Clearly seen that the

solution pH before aging gradually decreases on raising SO₃²⁻/Fe³⁺ ratio. Besides, increment of $SO_3^{2^-}/Fe^{3^+}$ ratio changes the color of the solution from light yellow to orange without forming any precipitates as shown in Fig. 5. These facts allow us to infer that added SO_3^{2-} alters the coordination of anions to Fe^{3+} in Fe^{3+} -complexes. To make clearer of this, UV spectra of the solution before aging were measured. The results are displayed in Fig. 6. At $SO_3^{2-}/Fe^{3+} = 0$, two absorption peaks appear around at 210 nm and 335 nm (spectrum a). The former and latter peaks are assignable to the coordination of OH⁻ and Cl⁻ to Fe³⁺ in Fe³⁺-complexes, respectively [25,26]. It should be noted that these peaks are weakened by raising SO₃²⁻/Fe³⁺ ratio (spectra **b** – **h**). Furthermore, at SO₃²⁻/Fe³⁺ \geq 0.15, new peak is found at ca. 305 nm and is identified as the coordination of SO_4^{2-} to Fe³⁺ in Fe³⁺-complexes (spectra **d** – **h**) [25,26]. It seems, therefore, that added SO_3^{2-} breaks the coordination of OH⁻ and Cl⁻ to Fe³⁺ in Fe³⁺-complexes. This can be explained by the difference of stability constant of Fe³⁺-complexes with anions. The logarithm of stability constants (log K) of Fe^{3+} -complexes with OH^{-} and CI^{-} are -2.9 and 0.6, respectively [22]. On the other hand, SO_{3}^{2-} forms a stable coordination to Fe³⁺ compared with OH⁻ and Cl⁻ [22]. Nevertheless, no peak characteristics due to coordination of SO_3^{2-} to Fe^{3+} can be found in all of the UV spectra. It is well-known that Na₂SO₃ is a strong reducing agent and is widely used as preservatives. Further, López-Miranda et al. reported the preparation of Ag nanoparticles in aqueous AgNO₃ solution using Na₂SO₃ as a reducing agent [27]. It seems, therefore, that Fe^{3+} in the solution before aging is reduced to Fe^{2+} by added SO_3^{2-} . To elucidate this, Fe^{2+} concentration ([Fe²⁺]) in the solution before aging was assayed by an *o*-phenanthroline color comparison method. Also, Fe^{3+} concentration ([Fe³⁺]) was calculated by subtracting from 100 mmol/L of initial $[Fe^{3+}]$ to assayed $[Fe^{2+}]$. Fig. 7 plots the $[Fe^{2+}]$ and $[Fe^{3+}]$ in the solution before aging, respectively represented by open and filled circles,

against molar ratio SO_3^{2-}/Fe^{3+} . Note that raising SO_3^{2-}/Fe^{3+} ratio linearly increases the $[Fe^{2+}]$ and decreases the $[Fe^{3+}]$, being indicative of the reduction of Fe^{3+} to Fe^{2+} by addition of SO_3^{2-} . Further, amount of formed Fe^{2+} in the solution is about twice as large as amount of Fe^{3+} . This means that one SO_3^{2-} ion reduces two Fe^{3+} ions and almost all the added SO_3^{2-} reacts with Fe^{3+} .

From these results, we can suppose the influence of SO_3^{2-} on the formation of β -FeOOH rust. The formation of iron rust particles in acidic Fe³⁺ solution was reported as follows [23,24,28]. The hexa-aqua ferric ions are produced by the reaction of Fe³⁺ with H₂O via reaction (5).

 $Fe^{3+} + 6H_2O$ $[Fe(H_2O)_6]^{3+} ---(5)$

The $[Fe(H_2O)_6]^{3+}$ is hydrolyzed to form $[Fe(OH)_n(H_2O)_{6-n}]^{3+}$ (n = 1 - 6) via protolysis reaction (6).

 $[Fe(H_2O)_6]^{3^+} + nH_2O \qquad [Fe(OH)_n(H_2O)_{6^-n}]^{3^+} + nH_3O^+ ---(6)$ The >Fe-OH groups of the $[Fe(OH)_n(H_2O)_{6^-n}]^{3^+}$ are condensed via oxolation reaction (7).

Also, the >Fe-OH groups of the $[Fe(OH)_n(H_2O)_{6-n}]^{3+}$ react with >Fe-H₂O groups via olation reaction (8).

>Fe-OH + H₂O-Fe< >Fe-OH-Fe< + H₂O ---(8)

The reactions (5) – (8) continuously progress during aging to form various kinds of iron rust particles. UV measurement of the solution before aging at $SO_3^{2^-}/Fe^{3^+}$ = 0 found the coordination of OH⁻ and Cl⁻ to Fe³⁺ in Fe³⁺-complexes, meaning the formation of [Fe(OH)_nCl_m(H₂O)_{6-n-m}]³⁺ by reaction (9):

 $[Fe(H_2O)_6]^{3+} + nH_2O + mCl^- [Fe(OH)_nCl_m(H_2O)_{6-n-m}]^{3+} + nH_3O^+ + mH_2O ---(9).$ Therefore, the protolysis, oxolation and olation reactions of $[Fe(OH)_nCl_m(H_2O)_{6-n-m}]^{3+}$ proceed during aging to crystallize as Cl⁻-containing β -FeOOH. When the SO₃²⁻ is added in the solution before aging, the SO₃²⁻ breaks the coordination of OH⁻ and Cl⁻ to Fe³⁺ in $[Fe(OH)_nCl_m(H_2O)_{6-n-m}]^{3+}$ and reduces the Fe³⁺ to Fe²⁺ via reaction (10), because the stability constant of Fe³⁺-complex with SO₃²⁻ is much higher than that with Cl⁻ and OH⁻ [22]:

$$2Fe^{3+} + SO_3^{2-} + H_2O \longrightarrow 2Fe^{2+} + SO_4^{2-} + 2H^+ ---(10).$$

The reaction (10) falls the solution pH, corresponding to the change of solution pH before aging against SO_3^{2-}/Fe^{3+} ratio shown in Fig. 5. It has been established that decrease of solution pH lowers the hydrolysis rate of Fe^{3+} and oxidation rate of Fe^{2+} [29]. Therefore, the Fe^{2+} formed by reaction (10) is difficult to oxidize to Fe^{3+} . As a result, the addition of SO_3^{2-} remarkably suppresses the formation and crystallization of β -FeOOH rust and the effect is enhanced on elevating $SO_3^{2^-}/Fe^{3^+}$ ratio. Also, a slight amount of α -FeOOH particles is generated in addition to β -FeOOH ones by adding SO₃²⁻. As stated above, almost all the SO₃²⁻ is oxidized to SO_4^{2-} via reaction (10). It seems, therefore, that the α -FeOOH formation is related to the generated SO_4^{2-} because the stability constant of Fe^{3+} -complex with SO_4^{2-} (4.1) is much larger than that with Cl⁻ and OH⁻ [22]. Hence, the yielded SO_4^{2-} preferentially coordinates to Fe^{3+} to form stable Fe^{3+} -complex such as $[Fe(OH)_n(SO_4)_{m/2}(H_2O)_{6-n-m}]^{3+}$. It has been reported that formation of SO42--containing Schwertmannite (Fe8O8(OH)6(SO4)) is formed by aging the $[Fe(OH)_n(SO_4)_{m/2}(H_2O)_{6-n-m}]^{3+}$ in acidic solution and the material is transformed into α -FeOOH [19-21,30-34]. Accordingly, the Schwertmannite particles are initially formed by adding SO32- and are transformed into irregular-shaped α -FeOOH ones during aging.

From the above-mentioned results, it is indicative that the SO_3^{2-} generated from corrosive SO_X gas in a coastal industrial atmosphere strongly suppresses the formation and crystallization of β -FeOOH rust on the steels.

4. Conclusions

From the information presented in this publication, following conclusions can be drawn. The yield of the products steeply decreased on elevating SO₃²⁻/Fe³⁺ ratio and reaches to zero at SO₃²⁻/Fe³⁺ ≥ 0.3. Also, increment of SO₃²⁻/Fe³⁺ ratio markedly lowered the crystallite size of β-FeOOH. These results indicate the suppression of formation and crystallization of β-FeOOH rust by adding SO₃²⁻. This could be identified as the change in coordination of anions to Fe³⁺ in Fe³⁺-complexes in the solution before aging by addition of SO₃²⁻. At SO₃²⁻/Fe³⁺ = 0, the coordination of OH⁻ and Cl⁻ to Fe³⁺ in Fe³⁺-complexes was recognized. Added SO₃²⁻ broke the coordination of OH⁻ and Cl⁻ to Fe³⁺ because the coordination of SO₃²⁻ to Fe³⁺ in Fe³⁺-complex is more stable than that of Cl⁻ and OH⁻. Further, added SO₃²⁻ reduces the Fe³⁺ to Fe²⁺ as following reaction: 2Fe³⁺ + SO₃²⁻ + H₂O --> 2Fe²⁺ + SO₄²⁻ + 2H⁺. As a result, [Fe³⁺] in the solution and solution pH were decreased, leading to the inhibition of formation and crystallization of β-FeOOH rust particles.

The obtained results infer that SO_3^{2-} produced by dissolution of corrosive SO_X gas in thin film water on the steels at a coastal industrial zone markedly suppresses the formation and crystallization of β -FeOOH rust.

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Fig. 1. Plots of yield of the β -FeOOH prepared at 85°C for 24 h as a function of molar ratio SO₃²⁻/Fe³⁺.



Fig. 2. XRD patterns of β-FeOOH prepared at different $SO_3^{2^-}/Fe^{3+}$ ratios and 85°C for 24 h. $SO_3^{2^-}/Fe^{3+}$ ratio: (a) 0, (b) 0.05, (c) 0.1, (d) 0.15, (e) 0.2, (f) 0.25.



Fig. 3. Plots of crystallite size of β -FeOOH prepared at 85°C for 24 h against molar ratio SO₃²⁻/Fe³⁺.



200 nm

Fig. 4. TEM pictures of β -FeOOH particles prepared at different SO₃²⁻/Fe³⁺ ratios and 85°C for 24 h. SO₃²⁻/Fe³⁺ ratio: (a) 0, (b) 0.05, (c) 0.1, (d) 0.15, (e) 0.2, (f) 0.25.



Fig. 5. (A) Picture of the solution before aging at different SO₃²⁻/Fe³⁺
 (B) plots of solution pH before aging as a function of molar ratio SO₃²⁻/Fe³⁺.



Fig. 6. UV spectra of the solution before aging at different $SO_3^{2^-}/Fe^{3_+}$ ratios. $SO_3^{2^-}/Fe^{3_+}$ ratio: (a) 0, (b) 0.05, (c) 0.1, (d) 0.15, (e) 0.2, (f) 0.25, (g) 0.3, (h) 0.4.



Fig. 7. Plots of (\bigcirc) [Fe²⁺] and (\bigcirc) [Fe³⁺] in the solution before aging as a function of molar ratio SO₃²⁻/Fe³⁺.