Mem. Fac. Sci. Shimane Univ., 23, pp. 85-88 Dec. 20, 1989

# Characterization of Vapour-phase Growth SnO<sub>2</sub> Single Crystal

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Since crystals of  $SnO_2$  were prepared by vapour-phase reaction method from anhydrous  $SnI_4$  powder. The single crystals were characterized by optical microscope and X-ray Lang camera in order to clear the growth mechanism and internal defects in them. A lot of black dots were observed in the Lang topograph. It seems that the dots are planar defects with the Magnéli structure in the crystal. The single crystals were grown by the Kossel mechanism.

## §1. Introduction

It is well-known that stannic oxide  $(SnO_2)$  with the rutile structure is a clear conducting material and has been widely used as clear electrodes. Many investigations have been made on a vapour-phase growth  $SnO_2$  single crystal.<sup>1-5)</sup> However most of the investigations have dealt with the growth condition or the morphorogy. There are little investigations on the growth mechanism or the internal defects. So, we have made  $SnO_2$  single crystals from the stannic halides with lower temperature vapour growth, which is preferable for avoiding contamination of the grown crystals.  $SnO_2$ single crystals have been characterized by optical microscope and X-ray diffraction topography technique of Lang. From the results, the growth mechanism and the internal defects will be discussed in this paper.

## §2. Experimantal Procedure

SnO<sub>2</sub> single crystals were grown by following vapour-phase reaction,

$$SnI_4 + O_2 \rightarrow SnO_2 + 2I_2$$
.

The arrangement used in this experiment is shown schematically in Fig. 1. Anhydrous  $SnI_4$  powder as the starting materials was placed at central part of furnace I, where the temperature was about 200°C. The  $SnI_4$  vapour was transported by  $N_2$  gas through a combustion tube, 20 mm in internal diameter, to the growth region in furnace II, whose temperature was varies between 100°C and 1300°C at each growth run. Dried O<sub>2</sub> gas



Sn I₄ + O₂ → Sn O₂ + 2 I₂

Fig. 1. Schematic illustration of arrengement used for growth SnO<sub>2</sub> single crystal.

or wet  $N_2$  gas was supplied through the glass pipe as the reaction gas. The flow rates of the dried  $N_2$  gas for carrying the  $SnI_4$  vapour and the dried  $O_2$  or wet  $N_2$  gas were 230 ml/min and 150ml/min, respectively. After growth runs of several hours, many crystals were observed on the tip of the combustion tube. They were identified as  $SnO_2$  crystals by X-ray diffraction. The growth condition and the morphorogies of  $SnO_2$  single crystals have been reported previously.<sup>6</sup>

A optical microscope was used for the observation of crystal surface. The internal defects in the crystal was observed by X-ray Lang camera. A characteristic X-ray used in this work was  $AgK_{\alpha 1}$  radiation. The accelerating voltage and the beam current were 50 kV and 1.5 mA, respectively. The diffraction plane used for the topograph was a crystallographic plane (200) and the exposure time was about 20 hours.  $AgK_{\alpha 1}$  radiation was exposed on Illford L-4 nuclear plate with emulsion thickness 50  $\mu$ m.

### §3. Experimental Results and Discussion

The crystallographic property of the  $SnO_2$  single crystal used in the present work is shown schematically in Fig. 2. The microphotograph of (011) plane is shown in Fig. 3. Many dots and lines are observed in this crystal surface. So, the microphotograph with high magnification is shown in Fig. 4. From this microphotograph, most of the dots and the lines are identified as the pits and the growth bands on the surface, respectively. A few dots are hillocks on the surface. Figure 5 shows the reverse plane of the crystal shown in Fig. 3 and many pits and growth bands are also observed.

Figure 6 is X-ray Lang topograph of the single crystal. In this figure, dislcation lines are not observed but a lot of black dots, whose mean diameter is about 20  $\mu$ m, are observed. These black dots don't correspond with the pits or hillocks on the crystal surface observed by the microscope. So they are lattice defects contained in the crystal. They may not be inclusions because the starting meterials and the carrier gas



Fig. 2. Crystallographic property of SnO<sub>2</sub> single crystal.



Fig. 3. Optical microphotograph of  $SnO_2$  single crystal. Growth bands and dots on the surface are observed.



Fig. 4. Optical microphotograph with high magnification of  $SnO_2$ . Inclined lines are growth bands or growth stepts. Most of dots are pits on the surface and a few hillocks are also observed.



Fig. 5. Optical microphotograph with reverse side of Fig. 3.



Fig. 6. X-ray topograph (AgK<sub>a1</sub>) of SnO<sub>2</sub> single crystal. A lot of black dots are observed.

are high purity. Kaito *et al.* suggested that  $SnO_2$  with the rutile sturucture prefers the Magnéri structure under the tin rich condition.<sup>7)</sup>. Therefore it seems that the black dot is a planar defect which has the Magnéri sturucture because this single crystal has been made by the vapour growth. The inclined thick lines are also observed at the central part of the crystal in the topograph. These lines correspond with the growth band observed in the microphotograph. The spiral pattern is not observed on the crystal surface and this crystal is dislocation free one. Therefore,  $SnO_2$  single crystal was grown by the Kossel mechanism and the surface steps are growth steps by the two dimensional nucleation mechanism.

In conclusion, the vapour-phase growth  $SnO_2$  single crystal is dislocation free and black dot defects in the crystal may be the planar defects with the Magnéri structure. Furthermore, this single crystal was grown by the Kossel mechanism.

#### References

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