Simulating study of atmospheric corrosion of steels in a coastal industrial zone: Effect of $SO_3^{2^-}$ on the formation of β -FeOOH rust particles synthesized from FeCl₃ solutions

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Abstract

For simulation of the atmospheric corrosion of steel in a coastal industrial district, β -FeOOH rusts were synthesized by aging the aqueous FeCl₃ solutions containing SO₃²⁻. The yield and crystallite size of β -FeOOH were remarkably decreased by adding SO₃²⁻, implying the suppression of formation and crystallization of β -FeOOH. Before aging, added SO₃²⁻ broke the coordination of Cl⁻ and OH⁻ to Fe³⁺ in Fe³⁺-complexes and reduced the Fe³⁺ to Fe²⁺. These facts suppose that SO₃²⁻ generated by dissolution of SO_X gas in thin film water on the steels at a coastal industrial zone markedly inhibits the formation of β -FeOOH rust.

Keywords: A. Steel; A. Iron; B. X-ray diffraction; C. Atmospheric corrosion; C. Rust

1. Introduction

Atmospheric corrosion of steels forms various kinds of steel rusts such as iron oxyhydroxides (α -, β - and γ -FeOOH), Fe₃O₄, poorly crystallized iron oxides and so forth. Among them, structure and composition of FeOOH rusts are strongly dependent of the exposure environment of steel [1-4]. The β -FeOOH rust is formed in Cl⁻-containing environment such as coastal and marine zones [1-2]. While, the α - and γ -FeOOH rusts are generated in urban and industrial zones including SO_x and NO_x in atmosphere [4]. This difference can be explained by considering that the Cl⁻, SO_4^{2-} and NO_3^{-} respectively produced by dissolution of air-borne salt, SO_X and NO_X in thin film water on the steels markedly affect on the formation and composition of steel rusts [1-3]. Therefore, studying the influence of anions on the formation and structure of steel rusts is necessary to understand the atmospheric corrosion mechanism of steels. In particular, investigation of formation of β-FeOOH rust in the presence of anions generated from SO_X and NO_X is considerably important because major industrial districts of the world are developed at a coastal area. It is well-known that major components of SO_x and NO_x are SO_2 and NO_2 , respectively, and dissolution of these gases in thin film water on the steels leads to the following reactions [3]:

The corrosion behavior and rust formation of steel have been examined by exposure test in marine atmosphere containing SO₂ [5-12]. However, a little study has been done about the influence of SO₄²⁻ and NO₃⁻ on the formation of β -FeOOH rust [13-21]. Oh et al. indicated by preparation of rust particles in a mixture of FeCl₂ and FeSO₄ solutions that increasing molar ratio [SO₄²⁻]/[Cl⁻]

enhances the α-FeOOH formation [14]. Ishikawa et al. and Kamimura et al. reported that the formation and crystallization of β-FeOOH particles are suppressed by addition of SO₄²⁻ and this behavior is associated with the stability of coordination of anions to Fe³⁺ in Fe³⁺-complexes [16,17]. Also, the authors synthesized the steel rust particles in a mixture of aqueous FeCl₃, Fe₂(SO₄)₃ and Fe(NO₃)₃ solutions and revealed that the SO₄²⁻ markedly inhibits the β-FeOOH formation, while no remarkable effect of NO₃⁻ is recognized [19]. Similar results were found for the steel rust particles prepared by aerial oxidation of a mixture of aqueous FeCl₂, FeSO₄ and NaNO₃ solutions [20,21]. On the other hand, sulfite ions (SO₃²⁻) generated by reaction (2) have almost not received attention in rust formation because the SO₃²⁻ is finally oxidized to SO₄²⁻ via reaction (3). Nonetheless, since the stability of coordination of SO₃²⁻ to Fe³⁺ in Fe³⁺-complexes is higher than that of SO₄²⁻ [22], SO₃²⁻ would affect on the formation of β-FeOOH rust as well as SO₄²⁻.

The aim of this study was to clarify the effect of SO_3^{2-} on the formation and structure of β -FeOOH rust. So that, the β -FeOOH particles were synthesized by aging the aqueous FeCl₃ solutions containing different amounts of Na₂SO₃, and the formation and structure of the products were examined by various means. The results obtained must serve to elucidate the atmospheric corrosion mechanism of steels in a coastal industrial zone.

2. Experimental

2.1 Synthesis of β -FeOOH particles in the presence of SO₃²⁻

The β -FeOOH particles were synthesized by hydrolysis of aqueous FeCl₃ solutions as follows. Aqueous solutions (250 mL) dissolving 0.03 mol of FeCl₃ were prepared in a sealed polypropylene vessel. To the FeCl₃ solutions, aqueous solutions (50 mL) dissolving various amounts of Na₂SO₃ were added.

Then, molar ratio SO_3^{2-}/Fe^{3+} in the solution ranged from 0 to 0.4 and the Fe³⁺ concentration was 100 mmol/L. The solutions thus prepared were aged at 85°C for 24 h in an air oven without stirring. The resulting precipitates were filtered off using 0.45 µm Millipore filter, fully washed with deionized-distilled water and finally dried in an air oven at 50°C for 24 h. All the chemicals purchased from Wako Pure Chemical Co. were regent grade and used without further purification.

2.2 Characterization

The samples thus obtained were characterized by a variety of conventional techniques. Powder X-ray diffraction (XRD) patterns were taken by a Rigaku diffractometer with a Ni-filtered Cu*K* α radiation at 30 kV and 15 mA. Particle morphology was observed by a TOPCON transmission electron microscope (TEM) at 200 kV. UV spectra were measured by a Shimadzu UV-vis spectrometer and quartz cell. Fe²⁺ concentration in the solution was assayed by an *o*-phenanthroline color comparison method.

3. Results and Discussion

Fig. 1 plots yield of the product as a function of molar ratio $SO_3^{2^-}/Fe^{3^+}$. It is clearly seen that yield of the product linearly decreases with the increase of $SO_3^{2^-}/Fe^{3^+}$ ratio and reaches to zero at $SO_3^{2^-}/Fe^{3^+} \ge 0.3$. This fact indicates that added $SO_3^{2^-}$ strongly suppresses the formation of β -FeOOH rusts.

Fig. 2 shows the XRD patterns of the products prepared at different SO_3^{2-}/Fe^{3+} ratios. At $SO_3^{2-}/Fe^{3+} = 0$, the diffraction peaks due to β -FeOOH (PDF no. 34-1266) mainly develop at $2\theta = 11.8^{\circ}$, 16.8° , 26.6° , 34.1° , 35.1° , 39.0° , 46.3° and 55.9° , corresponding to the reflection from (101), (200), (301), (004), (211), (310), (114) and (215) planes of the crystal (pattern **a**). Besides, the peaks

related to secondary phase are not detected. On raising SO₃²⁻/Fe³⁺ ratio, the peaks characteristics of β -FeOOH are gradually weakened and broadened, while the position of the peaks is unchanged (patterns **b** - **f**). Furthermore, weak peaks due to α -FeOOH (no. 81-464) appear at SO₃²⁻/Fe³⁺ = 0.15 and are intensified with an increment of SO₃²⁻/Fe³⁺ ratio (patterns **d** - **f**). Fig. 3 plots the crystallite size of β -FeOOH evaluated from full width at half maximum of the (301) peak at 2 θ = 26.6° using the Scherrer equation against molar ratio SO₃²⁻/Fe³⁺. The crystallize size is 106 nm at SO₃²⁻/Fe³⁺ = 0 and is dramatically decreased by raising SO₃²⁻/Fe³⁺ ratio. From these results, it is most likely that formation and crystallization of β -FeOOH are markedly impeded by addition of SO₃²⁻.

Fig. 4 shows the TEM pictures of the particles synthesized at various molar ratios $SO_3^{2^-}/Fe^{3^+}$. The rod-shaped β -FeOOH particles are observed at $SO_3^{2^-}/Fe^{3^+}$ = 0 and their mean particle length and width are 278 nm and 61 nm, respectively (picture **a**). The size of β -FeOOH particles slightly decreases as $SO_3^{2^-}/Fe^{3^+}$ ratio increases to 0.1, meaning the suppression of growth of the particles by added $SO_3^{2^-}$ (pictures **b** and **c**). At $SO_3^{2^-}/Fe^{3^+} \ge 0.15$, the size of β -FeOOH particles increases and the irregular particles with a size of ca. 20 nm are formed (pictures **d** - **f**). The irregular particles would be identified as α -FeOOH because the formation of this material is recognized at the same $SO_3^{2^-}/Fe^{3^+}$ ratio by XRD measurements shown in Fig. 2.

The foregoing results indicate that the SO_3^{2-} generated by dissolution of SO_X gas in thin film water on the steels at a coastal industrial atmosphere remarkably inhibits the formation and crystallization of β -FeOOH rust.

It is well-known that formation of iron oxyhydroxide particles such as β -FeOOH is dependent of the solution pH [1,4,23,24]. Hence, pH of the solution before aging is plotted as a function of SO₃²⁻/Fe³⁺ ratio in Fig. 5. Clearly seen that the

solution pH before aging gradually decreases on raising SO₃²⁻/Fe³⁺ ratio. Besides, increment of $SO_3^{2^-}/Fe^{3^+}$ ratio changes the color of the solution from light yellow to orange without forming any precipitates as shown in Fig. 5. These facts allow us to infer that added SO_3^{2-} alters the coordination of anions to Fe^{3+} in Fe^{3+} -complexes. To make clearer of this, UV spectra of the solution before aging were measured. The results are displayed in Fig. 6. At $SO_3^{2-}/Fe^{3+} = 0$, two absorption peaks appear around at 210 nm and 335 nm (spectrum a). The former and latter peaks are assignable to the coordination of OH⁻ and Cl⁻ to Fe³⁺ in Fe³⁺-complexes, respectively [25,26]. It should be noted that these peaks are weakened by raising SO₃²⁻/Fe³⁺ ratio (spectra **b** – **h**). Furthermore, at SO₃²⁻/Fe³⁺ \geq 0.15, new peak is found at ca. 305 nm and is identified as the coordination of SO_4^{2-} to Fe³⁺ in Fe³⁺-complexes (spectra **d** – **h**) [25,26]. It seems, therefore, that added SO_3^{2-} breaks the coordination of OH⁻ and Cl⁻ to Fe³⁺ in Fe³⁺-complexes. This can be explained by the difference of stability constant of Fe³⁺-complexes with anions. The logarithm of stability constants (log K) of Fe^{3+} -complexes with OH^{-} and CI^{-} are -2.9 and 0.6, respectively [22]. On the other hand, SO_{3}^{2-} forms a stable coordination to Fe³⁺ compared with OH⁻ and Cl⁻ [22]. Nevertheless, no peak characteristics due to coordination of SO_3^{2-} to Fe^{3+} can be found in all of the UV spectra. It is well-known that Na₂SO₃ is a strong reducing agent and is widely used as preservatives. Further, López-Miranda et al. reported the preparation of Ag nanoparticles in aqueous AgNO₃ solution using Na₂SO₃ as a reducing agent [27]. It seems, therefore, that Fe^{3+} in the solution before aging is reduced to Fe^{2+} by added SO_3^{2-} . To elucidate this, Fe^{2+} concentration ([Fe²⁺]) in the solution before aging was assayed by an *o*-phenanthroline color comparison method. Also, Fe^{3+} concentration ([Fe³⁺]) was calculated by subtracting from 100 mmol/L of initial $[Fe^{3+}]$ to assayed $[Fe^{2+}]$. Fig. 7 plots the $[Fe^{2+}]$ and $[Fe^{3+}]$ in the solution before aging, respectively represented by open and filled circles,

against molar ratio SO_3^{2-}/Fe^{3+} . Note that raising SO_3^{2-}/Fe^{3+} ratio linearly increases the $[Fe^{2+}]$ and decreases the $[Fe^{3+}]$, being indicative of the reduction of Fe^{3+} to Fe^{2+} by addition of SO_3^{2-} . Further, amount of formed Fe^{2+} in the solution is about twice as large as amount of Fe^{3+} . This means that one SO_3^{2-} ion reduces two Fe^{3+} ions and almost all the added SO_3^{2-} reacts with Fe^{3+} .

From these results, we can suppose the influence of SO_3^{2-} on the formation of β -FeOOH rust. The formation of iron rust particles in acidic Fe³⁺ solution was reported as follows [23,24,28]. The hexa-aqua ferric ions are produced by the reaction of Fe³⁺ with H₂O via reaction (5).

 $Fe^{3+} + 6H_2O$ [Fe(H₂O)₆]³⁺ ---(5)

The $[Fe(H_2O)_6]^{3+}$ is hydrolyzed to form $[Fe(OH)_n(H_2O)_{6-n}]^{3+}$ (n = 1 - 6) via protolysis reaction (6).

 $[Fe(H_2O)_6]^{3^+} + nH_2O \qquad [Fe(OH)_n(H_2O)_{6^-n}]^{3^+} + nH_3O^+ ---(6)$ The >Fe-OH groups of the $[Fe(OH)_n(H_2O)_{6^-n}]^{3^+}$ are condensed via oxolation reaction (7).

Also, the >Fe-OH groups of the $[Fe(OH)_n(H_2O)_{6-n}]^{3+}$ react with >Fe-H₂O groups via olation reaction (8).

>Fe-OH + H₂O-Fe< >Fe-OH-Fe< + H₂O ---(8)

The reactions (5) – (8) continuously progress during aging to form various kinds of iron rust particles. UV measurement of the solution before aging at $SO_3^{2^-}/Fe^{3^+}$ = 0 found the coordination of OH⁻ and Cl⁻ to Fe³⁺ in Fe³⁺-complexes, meaning the formation of [Fe(OH)_nCl_m(H₂O)_{6-n-m}]³⁺ by reaction (9):

 $[Fe(H_2O)_6]^{3+} + nH_2O + mCl^{-} [Fe(OH)_nCl_m(H_2O)_{6-n-m}]^{3+} + nH_3O^{+} + mH_2O ---(9).$ Therefore, the protolysis, oxolation and olation reactions of $[Fe(OH)_nCl_m(H_2O)_{6-n-m}]^{3+}$ proceed during aging to crystallize as Cl⁻-containing β -FeOOH. When the SO₃²⁻ is added in the solution before aging, the SO₃²⁻ breaks the coordination of OH⁻ and Cl⁻ to Fe³⁺ in $[Fe(OH)_nCl_m(H_2O)_{6-n-m}]^{3+}$ and reduces the Fe³⁺ to Fe²⁺ via reaction (10), because the stability constant of Fe³⁺-complex with SO₃²⁻ is much higher than that with Cl⁻ and OH⁻ [22]:

$$2Fe^{3+} + SO_3^{2-} + H_2O \longrightarrow 2Fe^{2+} + SO_4^{2-} + 2H^+ ---(10).$$

The reaction (10) falls the solution pH, corresponding to the change of solution pH before aging against SO_3^{2-}/Fe^{3+} ratio shown in Fig. 5. It has been established that decrease of solution pH lowers the hydrolysis rate of Fe^{3+} and oxidation rate of Fe^{2+} [29]. Therefore, the Fe^{2+} formed by reaction (10) is difficult to oxidize to Fe^{3+} . As a result, the addition of SO_3^{2-} remarkably suppresses the formation and crystallization of β -FeOOH rust and the effect is enhanced on elevating $SO_3^{2^-}/Fe^{3^+}$ ratio. Also, a slight amount of α -FeOOH particles is generated in addition to β -FeOOH ones by adding SO₃²⁻. As stated above, almost all the SO₃²⁻ is oxidized to SO_4^{2-} via reaction (10). It seems, therefore, that the α -FeOOH formation is related to the generated SO_4^{2-} because the stability constant of Fe^{3+} -complex with SO_4^{2-} (4.1) is much larger than that with Cl⁻ and OH⁻ [22]. Hence, the yielded SO_4^{2-} preferentially coordinates to Fe^{3+} to form stable Fe^{3+} -complex such as $[Fe(OH)_n(SO_4)_{m/2}(H_2O)_{6-n-m}]^{3+}$. It has been reported that formation of SO42--containing Schwertmannite (Fe8O8(OH)6(SO4)) is formed by aging the $[Fe(OH)_n(SO_4)_{m/2}(H_2O)_{6-n-m}]^{3+}$ in acidic solution and the material is transformed into α -FeOOH [19-21,30-34]. Accordingly, the Schwertmannite particles are initially formed by adding SO32- and are transformed into irregular-shaped α -FeOOH ones during aging.

From the above-mentioned results, it is indicative that the SO_3^{2-} generated from corrosive SO_X gas in a coastal industrial atmosphere strongly suppresses the formation and crystallization of β -FeOOH rust on the steels.

4. Conclusions

From the information presented in this publication, following conclusions can be drawn. The yield of the products steeply decreased on elevating SO₃²⁻/Fe³⁺ ratio and reaches to zero at SO₃²⁻/Fe³⁺ ≥ 0.3. Also, increment of SO₃²⁻/Fe³⁺ ratio markedly lowered the crystallite size of β-FeOOH. These results indicate the suppression of formation and crystallization of β-FeOOH rust by adding SO₃²⁻. This could be identified as the change in coordination of anions to Fe³⁺ in Fe³⁺-complexes in the solution before aging by addition of SO₃²⁻. At SO₃²⁻/Fe³⁺ = 0, the coordination of OH⁻ and Cl⁻ to Fe³⁺ in Fe³⁺-complexes was recognized. Added SO₃²⁻ broke the coordination of OH⁻ and Cl⁻ to Fe³⁺ because the coordination of SO₃²⁻ to Fe³⁺ in Fe³⁺-complex is more stable than that of Cl⁻ and OH⁻. Further, added SO₃²⁻ reduces the Fe³⁺ to Fe²⁺ as following reaction: 2Fe³⁺ + SO₃²⁻ + H₂O --> 2Fe²⁺ + SO₄²⁻ + 2H⁺. As a result, [Fe³⁺] in the solution and solution pH were decreased, leading to the inhibition of formation and crystallization of β-FeOOH rust particles.

The obtained results infer that SO_3^{2-} produced by dissolution of corrosive SO_X gas in thin film water on the steels at a coastal industrial zone markedly suppresses the formation and crystallization of β -FeOOH rust.

Acknowledgements

The authors grateful thank to Mr. Tsunao Yoneyama of Department of Biosignaling and Radioisotope Experiment Center for Integrated Research in Science of Shimane University for his help with TEM observation. This study was partly supported by the Grant-in-Aid for Young Scientists Fund (B) (23760699) and Scientific Research (C) (26420739).

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Fig. 1. Plots of yield of the β -FeOOH prepared at 85°C for 24 h as a function of molar ratio SO₃²⁻/Fe³⁺.



Fig. 2. XRD patterns of β-FeOOH prepared at different $SO_3^{2^-}/Fe^{3+}$ ratios and 85°C for 24 h. $SO_3^{2^-}/Fe^{3+}$ ratio: (a) 0, (b) 0.05, (c) 0.1, (d) 0.15, (e) 0.2, (f) 0.25.



Fig. 3. Plots of crystallite size of β -FeOOH prepared at 85°C for 24 h against molar ratio SO₃²⁻/Fe³⁺.



200 nm

Fig. 4. TEM pictures of β -FeOOH particles prepared at different SO₃²⁻/Fe³⁺ ratios and 85°C for 24 h. SO₃²⁻/Fe³⁺ ratio: (a) 0, (b) 0.05, (c) 0.1, (d) 0.15, (e) 0.2, (f) 0.25.



Fig. 5. (A) Picture of the solution before aging at different SO₃²⁻/Fe³⁺
 (B) plots of solution pH before aging as a function of molar ratio SO₃²⁻/Fe³⁺.



Fig. 6. UV spectra of the solution before aging at different $SO_3^{2^-}/Fe^{3_+}$ ratios. $SO_3^{2^-}/Fe^{3_+}$ ratio: (a) 0, (b) 0.05, (c) 0.1, (d) 0.15, (e) 0.2, (f) 0.25, (g) 0.3, (h) 0.4.



Fig. 7. Plots of (\bigcirc) [Fe²⁺] and (\bigcirc) [Fe³⁺] in the solution before aging as a function of molar ratio SO₃²⁻/Fe³⁺.